Nonlinear Langevin equation (NLE) theory and its application to colloidal polymer glass

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Center for Soft Condensed Matter Physics & Interdisciplinary Research



Background

- The basics of the NLE theory
- The unique role of bond length in colloidal polymer glass





Backgroud

The basics of the NLE theory

The unique role of bond length in colloidal polymer glass







🐠 Glass transition





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MLE theory for colloidal glass

Motion of a single particle: r(t)--scalar

DDFT, Local Equilibrium Approximation, Vineyard Closure

$$\int \int \frac{\partial r(t)}{\partial t} = -\frac{\partial}{\partial r} F_{dyn} [r(t)] + \eta(t) + \eta(t) + \eta(t) + \eta(t)$$
short time friction white noise

$$\int \int \frac{\partial r(t)}{\partial t} = -\frac{\partial}{\partial r} F_{dyn} [r(t)] \qquad \alpha \equiv 3/2r_{loc}^2 \qquad \text{Naive MCT}$$

$$Kirkpatrick \& Wolynes, 1987 \qquad \alpha = \frac{1}{2}\beta K(t \to \infty)$$

$$\int \int \frac{\partial q}{\partial r} F_{dyn} [r(t)] \qquad \qquad = \frac{1}{6} \int \frac{\partial q}{(2\pi)^3} \rho q^2 C^2(q) S(q) e^{-q^2/4\alpha(1+S^{-1}(q))}$$

MLE theory for colloidal glass

Motion of a single particle: r(t)--scalar

DDFT, Local Equilibrium Approximation, Vineyard Closure

$$\int \int \frac{\partial r(t)}{\partial t} = -\frac{\partial}{\partial r} F_{dyn} [r(t)] + \eta(t)] \quad \text{K.S. Schweizer J. Chem. Phys. (2005)}$$

short time friction white noise

$$F_{dyn}(r) = -3k_BT \ln(r) - k_BT \int \frac{d\bar{q}}{(2\pi)^3} \rho C^2(q) S(q) [1 + S^{-1}(q)]^{-1} e^{-q^2 r^2 [1 + S^{-1}(q)]/6}$$

translational entropy Inter-particle interactions (caging effect)
favors fluid localized solid

S(q): Static Structure Factor C(q): Direct Correlation Function





> Crossover: Φ

Localized, activated hopping dynamics

 \succ Characteristic lengths: r_{LOC} , R^* , r_{R}

The "confined" space of the particle's motion- "cage size"

Related to the Lindemann length at the crossover

 \blacktriangleright Barrier and the relaxation time: F_{B} , τ_{α}

Structural relaxation

Escaping the cage





Mirigian and Schweizer JPCL 2013



$$F_{tot} = F_B + F_e$$

Kramers first-passage theory:

$$\tau_{\alpha} = \tau(\phi) \exp(F_{tot}/k_B T)$$

For hard-sphere

$$\frac{\tau(\phi)}{\tau_0} = \frac{2\pi g(\sigma) k_B T}{\sigma^2 \sqrt{K_0 K_B}}$$

$$\tau_0 = \sigma^2 \zeta_0 / k_B T$$

 $au(\phi)$ is a weak function of volume fraction



 $\phi_c \text{ or } 1/T_c$







Backgroud

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Yang et al., Phys. Chem. Chem. Phys. 2010

Zhao et al., J. Phys. Chem. Lett. 2013



Features of molecular polymer

- Bond length is not a controllable variable
- Ambiguity of the basic unit

Colloidal polymer as a model system

- Unambiguity of the basic unit
- Flexibility in tuning the architecture, especially the bond length

<u>Current theoretical and simulation research</u>

- Based on Gaussian or bead-spring chain
- Focus on the influence of chain length and rigidity





NLE theory for chain molecules

$$F_{dyn}(r) = -3k_{B}T\ln(r) - k_{B}T\int \frac{d\bar{q}}{(2\pi)^{3}}\rho C^{2}(q)S(q) \left[1 + S^{-1}(q)\right]^{-1} e^{-q^{2}r^{2}\left[1 + S^{-1}(q)\right]/6}$$

$$F_{dyn} = -3k_{B}T\ln(r) - k_{B}T\int \frac{d\bar{q}}{(2\pi)^{3}} \left[C^{2}(q)\rho S(q)\omega(q) + C_{intra}(q)(\omega - 1)\right]$$

$$\times \left[1 + S^{-1}(q)\right]^{-1} e^{-q^{2}r^{2}\left[1 + S^{-1}(q)\right]/6}$$

$$\omega(q) = \frac{1}{(2\pi)^{3}} \left[C^{2}(q)\rho S(q)\omega(q) + C_{intra}(q)(\omega - 1)\right]$$

$$\omega(\mathbf{q}) = \frac{1}{1 - \rho C_{\text{intra}}(\mathbf{q})}$$



$$h(q) = \omega(q)C(q)[\omega(q) + \rho h(q)] \qquad S(q) = \omega(q) + \rho h(q)$$

Intramolecular correlation function

Koyama distribution for worm-like chain

l: bond length ε : rigidity *N*: chain length





$$\left\langle r_{ab}^{2} \right\rangle = \left| a - b \right| l^{2} \left[\frac{1 - \left\langle \cos \theta \right\rangle}{1 + \left\langle \cos \theta \right\rangle} + \frac{2 \left\langle \cos \theta \right\rangle}{n} \frac{1 - \left(- \left\langle \cos \theta \right\rangle \right)^{|a-b|}}{\left(1 + \left\langle \cos \theta \right\rangle\right)^{2}} \right]$$

$$\left\langle r_{ab}^{4} \right\rangle = \left\langle r_{ab}^{2} \right\rangle^{2} + D_{ab}l^{4} \quad D_{ab} \longrightarrow D_{ab} (\langle \cos \theta \rangle, \langle \cos^{2} \theta \rangle)$$

$$\left\langle \cos\theta\right\rangle = \frac{\int_{\theta_0}^{\pi} e^{-\varepsilon(1+\cos\theta)} \cos\theta \sin\theta d\theta}{\int_{\theta_0}^{\pi} e^{-\varepsilon(1+\cos\theta)} \sin\theta d\theta} = \frac{1}{\varepsilon} - \frac{e^{\varepsilon} + \cos\theta_0 e^{-\varepsilon\cos\theta_0}}{e^{\varepsilon} - e^{-\varepsilon\cos\theta_0}}$$

$$\left\langle \cos^2 \theta \right\rangle = \frac{2}{\varepsilon} \left\langle \cos \theta \right\rangle + \frac{e^{\varepsilon} - \cos^2 \theta_0 e^{-\varepsilon \cos \theta_0}}{e^{\varepsilon} - e^{-\varepsilon \cos \theta_0}}$$

Intramolecular correlation function



 $1+2(N-1)\sin(ql)/Nql$

Asymptotics between colloidal and Gaussian chain of short bond length







 $\phi = 0.55$



- *l*=4/3 has the largest barrier height.
- The barrier height increases with the increase of rigidity and decreases with chain length.



Unique role of bond length



Barrier and hopping time nonmonotonicity 25 25 *l*=1.1 l=4/3, ε=0, №=10 F_{tot} 1=4/3 hard sphere 25 *l*=4/3 *ε*=0, *N*=10 20 20 l = 1.7l = 1.120 hard sphere 15 $\frac{F_{tot}}{k_{\scriptscriptstyle B}T}^{\rm 15}$ 15 Barriers *l*=1.7 15 hard sphere 10 *ε*=0, *№*=10 10 5 5 5 0 0 0.50 0.52 0.54 0.56 0.58 0.60 0.62 0.15 0.2 0.05 0.10 0.00 £=0 25 l=4/3, N=10 $\phi - \phi_c$ ε**=0.4** Ø 20 *ε*=0.7 25 F_{tot} 15 25 *ε*=0 *№*=10 hard sphere =0.4 **№**=20 $k_{\rm B}T$ 10 20 20 =0.7 №=50 $\frac{F_{tot}}{k_BT}$ 15 F_{tot} 15 hard sphere *N*=100 $k_{\scriptscriptstyle B}T_{-10}$ hard sphere 10 25 N=10 *l*=4/3, ε=0 5 5 N=20 20 *l*=4/3, *N*=10 *l*=4/3, *ε*=0 N=50 0 0 15 N=100 0.09 0.12 0.05 0.10 0.15 0.06 0.15 0.18 0.2 0.00 hard sphere 10 $\phi - \phi_c$ $\phi - \phi_c$ 5 0.50 0.52 0.54 0.56 0.58 0.60 0.6

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Unique role of bond length in determining the barriers



$$\tau_{\alpha} \propto \exp\left(u_{0}^{2}/\langle u^{2} \rangle\right)$$

$$\tau_{\alpha} \propto \exp\left(a^{2}/2\langle u^{2} \rangle + \delta^{2}/8\langle u^{2} \rangle^{2}\right)$$

$$\tau_{\alpha} \propto \exp\left[\left(u_{0}^{2}/\langle u^{2} \rangle\right)^{\alpha/2}\right]$$

PNAS 112, 2966 (2015)





Unique role of bond length in the universal relationship between localization length and relaxation time.

nonmonotonicity



- Colloidal polymer is a promising model system to study glassy behavior of chain "molecules".
- The influence of chain length and rigidity on the local length scale and activated barrier is encoded in the critical volume fraction.
- In contrast, bond length plays a special role in determining the behavior of localization length and dynamic barrier.
- We find nonmonotonic dependences of structure factor, barrier height and the universal relationship on the bond length.



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Simulation as the input



Possible reasons for the discrepancy:

- 1. density-controlled vs temperature-controlled
- incomplete consideration of intrachain excludedvolume interaction



Mechanical response and Active glass

r

$$\overrightarrow{} = \overrightarrow{} = \overrightarrow{\phantom$$

 τ : Applied Stress

External force on segment $\propto \sigma^2 \tau$

Instantaneous Mechanical work ~ $-\tau \Delta V \propto -\sigma^2 \tau r$

$$F_{dyn}(r;\tau) = F_{dyn}(r;\tau=0) - c\sigma^2 \tau r$$

Structural correlations **NOT** affected by stress



Brownian motion + driven by propelling force

Modified structural correlations by simulation + the stress effect?



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